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# Electrolyte effects on photoelectron injection and recombination dynamics in dye-sensitized solar cells

Changneng Zhang, Zhipeng Huo, Yang Huang, Songyuan Dai\*, Meng Wang, Yingtong Tang, Yifeng Sui

Key Laboratory of Novel Thin Film Solar Cells, Institute of Plasma Physics, Chinese Academy of Sciences, P.O. Box 1126, Hefei, Anhui 230031, PR China

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#### ABSTRACT

In this study, the influence of electrolyte on the charge recombination, electron injection efficiency ( $\Phi_{inj}$ ) and performance of dye-sensitized solar cells (DSCs) was investigated. A negative shift in the flatband potential ( $V_{fb}$ ) of the TiO<sub>2</sub> electrode indicated that 1-methylbenzimidazole (MBI) chemisorbed on the TiO<sub>2</sub> surface and reduced the Li<sup>+</sup> adsorption in the nanostructured electrode. Electrochemical impedance spectra for the DSCs showed that the addition of MBI in the LiI electrolyte could increase the electron lifetime and reduce the rate constant ( $k_{et}$ ) for  $I_3^-$  reduction with electrons in the TiO<sub>2</sub> conduction band. The increase in  $V_{oc}$  for solar cells with MBI was mainly attributed to the negative shift in the  $V_{fb}$  of the TiO<sub>2</sub> electrody and the decrease of recombination reaction at the dyed TiO<sub>2</sub>/electrolyte interface. Analysis of the short-circuit current density ( $J_{sc}$ ) depended on the light intensity for dye-sensitized solar cells indicated that there was no obvious effect on  $\Phi_{inj}$  when the LiI concentration was up to 0.7 M in the electrolyte. Due to the negative shift in  $V_{fb}$  of the TiO<sub>2</sub> electrode by the addition of MBI in the electrolyte,  $\Phi_{inj}$  could be reduced to obtain a decreased  $J_{sc}$ .

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#### 1. Introduction

Dye-sensitized solar cells (DSCs) developed by Grätzel and co-workers have been extensively studied for their high-power conversion efficiency and potential low cost [1–8]. The typical DSC is normally made into sandwich-type photoelectrochemical solar cells between the dyed TiO<sub>2</sub> electrode and the platinized counter electrode, filling with the electrolyte commonly containing the  $I^-/I_3^-$  redox couple. Under illumination, photoexcited sensitizers inject electrons into the TiO<sub>2</sub> conduction band. The quantum yield of the photoelectron injection depends on the surface properties of TiO<sub>2</sub> and the nature and concentration of cations in the electrolyte [9]. The injected electrons can transfer to oxidized dye or  $I_3^-$  in the electrolyte resulting in the positive shift of the conduction band. Current strategy to improve the performance of DSCs is related to suppressing the charge recombination and increasing the effective electron injection efficiency.

It has been found that cations in the electrolyte could induce a shift in the flatband potential of  $TiO_2$  electrode and affect the rate of electron injection [10,11]. Grätzel reported that the flatband potential equals the Fermi level of the nanostructured  $TiO_2$  electrode, and is sensitive to the contact electrolyte [12]. Redmond and Fitz-

E-mail address: sydai@ipp.ac.cn (S. Dai).

maurice investigated cation effects on the flatband potential of the nanostructured  $TiO_2$  electrode by spectroscopic techniques [13]. They found that the increase of cation content in the electrolyte led to a positive shift in the flatband potential through intercalation into the crystal lattice or chemisorption on the surface of the  $TiO_2$  electrode. Meyer and co-workers reported that Li<sup>+</sup> could accelerate the rate of the electron injection from the photoexcited sensitizers into the  $TiO_2$  conduction band, resulting in a larger short-circuit photocurrent and a lower open-circuit voltage [10,11].

Nitrogen-containing heterocycle, such as 1-methylbenzimidazole (MBI) and 4-t-butylpyridine (tBP), is widely used due to the effective improvement on the open-circuit voltage of DSCs [14-20]. Schlichthorl et al. found that the addition of tBP in the electrolyte resulted in an increased electron lifetime and a significant upward shift of the conduction band edges [17]. Hagfeldt and co-workers found that the addition of tBP in the electrolyte resulted in a decreased response in the red part of the photocurrent spectrum [18,19]. They ascribed this effect to a negative shift in the  $TiO_2$ conduction band edge. With respect to interface property of the nanostructured TiO<sub>2</sub> electrode in the presence of MBI, we reported the negative shift of the flatband potential resulting in a larger increase in the open-circuit voltage. This result implies that MBI interaction with TiO2 could suppress Li<sup>+</sup> ions on the surface of nano-TiO<sub>2</sub> electrode [20]. However, there is no detailed study about the influence of the shift in the flatband potential of the nanostructured TiO<sub>2</sub> electrode on the photoelectron injection from the dye to the TiO<sub>2</sub> conduction band and the rate constant for the recombination reaction at the dyed TiO<sub>2</sub>/electrolyte interface

<sup>\*</sup> Corresponding author at: Key Laboratory of Novel Thin Film Solar Cells, Institute of Plasma Physics, Chinese Academy of Sciences, P.O. Box 1126, Hefei, Anhui 230031, PR China. Tel.: +86 551 5591377; fax: +86 551 5591377.

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Table 1

Composition and concentration of various electrolytes used in this study.

Electrolyte	DMPII, MBI, LiI
Α	0.7 M DMPII
В	0.6 M DMPII, 0.45 M MBI, 0.1 M LiI
С	0.6 M DMPII, 0.1 M LiI
D	0.4 M DMPII, 0.3 M LiI
E	0.7 M LiI
F	0.7 M DMPII, 0.45 M MBI

In this paper, we attempt to study the effect of the negative shift in the flatband potential ( $V_{\rm fb}$ ) of the nanostructured TiO<sub>2</sub> electrode on the rate constant ( $k_{\rm et}$ ) for the recombination reaction at the dyed TiO<sub>2</sub>/electrolyte interface. And we also study the electron injection efficiency ( $\Phi_{\rm inj}$ ) and photovoltaic performance of DSCs with MBI and Li<sup>+</sup> in the electrolyte.

#### 2. Experimental

#### 2.1. Materials

1-Methylbenzimidazole (MBI) and lithium perchlorate (LiClO<sub>4</sub>) were obtained from Aldrich. Anhydrous lithium iodide, iodine, and 3-methoxypropionitrile (MePN) were purchased from Fluka. Acetonitrile (MeCN) (chromatographic grade, Haoshen Chemical, Inc., Shanghai, China) was used without further purification. 1,2-Dimethyl-3-propylimidazolium iodide (DMPII) was synthesized as reported previously [15]. Six different electrolytes were employed for DSCs as shown in Table 1.

#### 2.2. DSC assembly

The colloidal TiO<sub>2</sub> nanoparticles were prepared by hydrolysis of titanium tetraisopropoxide as described elsewhere [20]. Nanocrystalline electrodes about 12  $\mu$ m in thickness were obtained by screen-printing TiO<sub>2</sub> paste on FTO glass (TEC-8, LOF). The TiO<sub>2</sub> film was heated in air for 30 min at 450 °C. The film was immersed in an ethanol solution of 0.5 mM cis-dithiocyanate-N,N'-bis-(4carboxylate-4-tetrabutylammoniumcarboxylate-2,2'-bipyridine) ruthenium (II) (N719) for 12 h. The platinized counter electrodes were obtained by spraying H<sub>2</sub>PtCl<sub>6</sub> solution to FTO glass followed by heating at 410 °C for 20 min. DSCs were fabricated by sealing the counter electrode and the dyed TiO<sub>2</sub> film with thermal adhesive films (Surlyn, Dupont). And the electrolyte was filled from a hole made on the counter electrode, which was later sealed by a cover glass and thermal adhesive films.

#### 2.3. Methods

The photovoltaic performances of DSCs with the active area of  $4.0 \text{ cm}^2$  were measured under an illumination of AM 1.5 ( $100 \text{ mW cm}^{-2}$ ), which was realized on a solar simulator (Changchun Institute of Optics Fine Mechanics and Physics, Chinese Academy of Science) with a Keithley 2420 source meter.

The flatband potential ( $V_{fb}$ ) of the nanostructured TiO<sub>2</sub> electrodes was performed by measuring absorbance at 780 nm as a function of the applied potential. For spectroscopic electrochemistry measurement, 4-µm thick TiO<sub>2</sub> film formed the working electrode (2 cm<sup>2</sup> surface area) of a three-electrode photoelectrochemical cell employing a platinum wire counter electrode and an Ag/AgCl reference electrode. Potential control was carried out on a CHI 660A potentiostat, the applied potential being scanned at 5 mV/s. A 780 nm monochromatic light source was obtained from UV–Vis spectrophotometer (TU-1901, PGeneral Instrument Inc., China). For each determination of  $V_{fb}$ , a new working electrode and freshly prepared electrolyte solution were used.



**Fig. 1.** Equivalent circuit used for the impedance spectra of DSCs.  $R_s$ : serial resistance;  $R_{Pt}$ : the resistance for charge transfer across the Pt/redox electrolyte interface,  $C_{Pt}$ : the chemical capacitance (expressed as a constant phase element) at the Pt/redox electrolyte interface,  $R_{ct}$ : the resistance for charge transfer across the TiO<sub>2</sub>/redox electrolyte interface,  $C_{\mu}$ : the chemical capacitance (expressed as a constant phase element) of the TiO<sub>2</sub> nanoparticles,  $Z_{d(sol)}$ : the ions diffusion impedance in the electrolyte.

Steady-state voltammograms were recorded on a CHI 660A electrochemical workstation at room temperature in two-electrode mode of DSCs. Impedance measurements were done using an IM6ex electrochemical workstation (Zahner-Elektrick, Germany) in the frequency range of 50 mHz to 1000 kHz at room temperature. The amplitude of the alternative signal was 5 mV. The impedance measurements were carried out in the dark and the obtained spectra were fitted with Z-View software in terms of appropriate equivalent circuit as shown in Fig. 1 [21].

#### 3. Results and discussion

## 3.1. Effect of additives (1-methylbenzimidazole and LiI) on $V_{fb}$ of the TiO<sub>2</sub> electrode

The role of MBI in the electrolyte in determining  $V_{\rm fb}$  of TiO<sub>2</sub> electrode was obtained by spectroelectrochemical techniques, measuring the absorbance at 780 nm as a function of the given potential shown in Fig. 2. The value of  $V_{\rm fb}$  determined for 0.5 M LiClO<sub>4</sub> in MeCN (-0.74V vs. Ag/AgCl) was 1.47V, which was in accordance with the literature [13]. The positive shift in  $V_{\rm fb}$  observed at 0.5 M LiClO<sub>4</sub> was attributed to the specific adsorption of Li<sup>+</sup> on the TiO<sub>2</sub> electrode. As shown in Fig. 2, the addition of MBI in the electrolyte with 0.5 M LiClO<sub>4</sub> produced a value for  $V_{\rm fb}$  of -0.98 V, which was 0.24 V more negative than that for LiClO<sub>4</sub> in MeCN. This negative shift could be attributed to MBI chemisorption with TiO<sub>2</sub> to reduce the adsorption of Li<sup>+</sup> on the electrode surface.



**Fig. 2.** Absorbance measured at 780 nm as a function of applied potential for a nanostructured  $TiO_2$  electrode in MeCN (0.5 M LiClO<sub>4</sub>) and with the addition of 0.5 M MBI. The calculated  $V_{\rm fb}$  values were obtained by extrapolating the linear portions of these plots, and the applied potential was scanned at 5 mV/s.



**Fig. 3.** Nyquist plots of DSCs with electrolytes A–C and F measured at -0.51 V in the dark. The lines show the fitted results. Inset represents enlargement of the area marked with circle.

#### 3.2. Effect of additives on interface recombination in DSCs

Fig. 3 shows the Nyquist plots of DSCs with electrolytes A-C and F measured at forward bias of -0.51 V. The small semicircle in the higher frequencies is due to the counter electrode and could be described by the redox charge transfer resistance  $(R_{\rm Pt})$ , and double layer capacitance (C<sub>Pt</sub>, expressed as constant phase element). The large semicircle at lower frequencies is ascribed to the electron accumulation and recombination processes in the TiO<sub>2</sub> layer, and chemical capacitance (expressed as constant phase element) [21–23]. Fitting the larger semicircle at lower frequency gives the chemical capacitance ( $C_{\mu}$ ) and the charge transport resistance ( $R_{ct}$ ) in the EIS experiment. The product of these two quantities gives a time constant,  $\tau = R_{ct}C_{\mu}$ , which corresponds to the observed electron lifetime  $(\tau)$  [23]. When LiI (0.1 M) was added to the 0.7 M DMPII electrolyte (electrolyte A), the chemical capacitance  $C_{\mu}$  increased from 2.8 mF to 24.3 mF and the charge transport resistance  $R_{ct}$ decreased sharply from 44.2  $\Omega$  to 3.5  $\Omega$ , yielding the observed  $\tau$  of 85.1 ms for DSCs with 0.1 M LiI electrolyte (electrolyte C) in comparison with that for DSCs without Lil (electrolyte A), i.e., 123.8 ms. But when MBI (0.45 M) was added to the 0.7 M DMPII electrolyte (electrolyte A), the chemical capacitance  $C_{\mu}$  decreased to 0.64 mF and the charge transport resistance  $R_{ct}$  increased to 586  $\Omega$ , yielding the observed  $\tau$  of 375 ms for DSCs without LiI in the electrolyte (electrolyte F). It is indicated that a decreased  $\tau$  for DSCs with 0.1 M Lil electrolyte (electrolyte C) revealed that Li<sup>+</sup> could shift the conduction band of the TiO<sub>2</sub> semiconductor downward and increase the charge recombination at the dyed TiO<sub>2</sub>/electrolyte interface. When MBI was introduced to electrolyte C, the chemical capacitance  $C_{\mu}$  decreased to 1.3 mF and the electron transport resistance  $R_{\rm ct}$  increased sharply to 452.4  $\Omega$ , yielding a longer electron lifetime of 588.1 ms for DSCs with MBI in the electrolyte (electrolyte B). It is clearly seen that the addition of MBI in the electrolyte could decrease the recombination reaction with  $I_3^-$ , yielding an increased  $\tau$  for solar cells.

To study the trap distribution in the TiO<sub>2</sub> electrode and the rate constant  $k_{\rm et}$  for charge recombination at the dyed TiO<sub>2</sub>/electrolyte interface, EIS was measured at different negative applied potentials for DSC with electrolytes A–C.  $C_{\mu}$  and  $R_{\rm ct}$  have been expressed in the following models [21,24–27].

$$C = C_{\rm a} \exp\left[\frac{-\alpha eV}{k_{\rm B}T}\right] + C_b \tag{1}$$

$$R_{\rm ct} = R_{\rm ct}^{\rm a} \exp\left[\frac{\alpha_{\rm t} eV}{k_{\rm B}T}\right] + R_{\rm ct}^{\rm b}$$
<sup>(2)</sup>

where  $k_{\rm B}$  is the Boltzmann constant, *T* is the temperature, *e* is the elementary charge, *V* is the applied potential,  $C_{\rm a}$  and  $R_{\rm ct}^{\rm a}$  are the



**Fig. 4.** Results from the impedance data for DSC with electrolytes A–C in the dark at the different applied potentials. (a) Capacitance of DSC with electrolytes A–C. (b) Charge transfer resistance of DSC with electrolytes A–C. The lines indicate the fitted results of  $C_{\mu}$  using Eq. (1) and  $R_{ct}$  using Eq. (2) as fitting models.

pre-exponential factor,  $C_{\rm b}$  is the quasiconstant capacitance at low potentials and  $R_{\rm ct}^{\rm b}$  is the lower charge transfer limit,  $\alpha$  is a coefficient describing an exponential distribution of trap states below the conduction band edge ( $\alpha < 1$ ),  $\alpha_{\rm t}$  is the transfer coefficient of the faradic reaction.

For a (pseudo) first-order recombination reaction with  $I_3^-$ , the reaction rate constant  $k_{\text{et}}$  can be calculated as expressed in Eq. (3) [21].

$$k_{\rm et} = A \cdot \exp\left(\frac{E_{\rm a}^0 + \alpha_{\rm t} \cdot F \cdot V}{RT}\right) \tag{3}$$

Here *A* is the pre-exponential factor,  $E_a^0$  is the activation energy at the potential of 0 V and independent of temperature, and *F* the Faraday constant. Eq. (3) implies that  $k_{et}$  depends on the parameter of  $\alpha_t$  and increases with the applied potential at room temperature.

Fig. 4 shows the fitted values of  $C_{\mu}$  using Eq. (1) and  $R_{ct}$  using Eq. (2) as fitting models. The fit yielded  $\alpha$  values from Eq. (1) were 0.29 for DSC without Lil (electrolyte A), 0.28 for DSCs with 0.1 M Lil electrolyte (electrolyte C), and 0.24 for DSC with MBI in the electrolyte (electrolyte B), which was in accordance with the literature [21,26]. This demonstrated that the trap distribution in energy below the TiO<sub>2</sub> conduction band was not affected in the presence of MBI. The fit yielded  $\alpha_t$  values from Eq. (2) were 0.48 for DSCs without Lil (electrolyte C), and 0.40 for DSCs with 0.1 M Lil electrolyte (electrolyte B). From Eq. (3), we concluded that MBI could decrease  $k_{et}$  to retard the recombination reaction in the TiO<sub>2</sub> electrode, which followed the sequence  $k_{et}$  (with Lil/MBI) <  $k_{et}$  (without Lil) <  $k_{et}$  (with Lil).



**Fig. 5.** Energy diagram for DSC with electrolyte containing  $I^-/I_3^-$  redox. The energy levels of nano-TiO<sub>2</sub>, dye, redox, the photoexcitation process, and the electron injection in the conduction band of nano-TiO<sub>2</sub> are shown in the diagram. The solid arrows indicate electron flowways, while the dashed arrows are the possible recombination pathways with the oxidized sensitizer and with the redox couple.

The predominant recombination in DSCs is expected to be the photoinjected electrons reaction with  $I_3^-$  ions, due to the fast regeneration of dye by  $I^-$  in the electrolyte. And  $V_{oc}$  for DSCs with  $I^-/I_3^-$  redox electrolyte is also determined by the following equation [2,28]:

$$V_{\rm oc} = \frac{kT}{e} \ln\left(\frac{I_{\rm inj}}{n_{\rm cb}k_{\rm et}[I_3^-]}\right) \tag{4}$$

where *k* and *T* are the Boltzmann constant and absolute temperature, respectively,  $I_{inj}$  is the injection electron from dye to semiconductor,  $n_{cb}$  the electron concentration at the conduction band edge of semiconductor, and  $k_{et}$  the rate constant for the back electron transfer from the conduction band of TiO<sub>2</sub> to  $I_3^-$  ions. According to Eq. (4),  $V_{oc}$  increases with the decreasing of  $k_{et}$ . So MBI chemisorption with TiO<sub>2</sub> to reduce the adsorption of Li<sup>+</sup> could improve the  $V_{oc}$  for solar cells.

The increased  $V_{oc}$  for DSCs in the presence of MBI compared with that in the absence of MBI may be explained by  $V_{fb}$  of the TiO<sub>2</sub> photoelectrode. Under Fermi level pinning, these two parameters are linked by the expression [28,29]

$$V_{\rm oc} = \left| V_{\rm fb} - V_{\rm red} \right| \tag{5}$$

where  $V_{red}$  is the standard reduction potential of a redox couple assuming that  $V_{red}$  does not vary with the addition of MBI. From the energy diagram of the nanocomposite depicted in Fig. 5 in our cell system, it can be seen that the increased  $V_{oc}$  for DSCs with the addition of MBI was due to the negative shift of  $V_{fb}$  with MBI chemisorption on TiO<sub>2</sub> photoelectrode to decrease the adsorption of Li<sup>+</sup> on the electrode surface. From Eqs. (4) and (5), we obtain

$$\left|V_{\rm fb} - V_{\rm red}\right| = \frac{kT}{e} \ln\left(\frac{I_{\rm inj}}{n_{\rm cb}k_{\rm et}[I_3^-]}\right) \tag{6}$$

From Eq. (6), it was found that the decrease of the rate constants  $k_{\rm et}$  for  $I_3^-$  reduction at the TiO<sub>2</sub>/electrolyte interface was attributed to the negative shift in  $V_{\rm fb}$  of the TiO<sub>2</sub> electrode. It was suggested that the addition of MBI in the electrolyte could rise up the  $V_{\rm fb}$  of the TiO<sub>2</sub> electrode and decrease the  $k_{\rm et}$  for  $I_3^-$  reduction at the TiO<sub>2</sub>/electrolyte interface.



**Fig. 6.** Short-circuit current  $J_{sc}$  as a function of light intensity for DSCs with electrolytes A–E. Inset represents the slopes for electrolytes A–E.

#### 3.3. Effect of additives on photoinjected electrons in DSCs

Fig. 6 shows that the  $J_{sc}$  of DSCs with electrolytes A–E varied in direct proportion to the illumination intensity (I) (AM 1.5, 46.4–100 mW cm<sup>-2</sup>). Because recombination at short circuit is negligible,  $J_{sc}$  could be related to the irradiation intensity *I* by the expression [30].

$$J_{\rm SC} = q\phi A I \tag{7}$$

where *q* is the electronic charge, *A* is a constant and depends on the light harvesting efficiency of dye N719 at each wavelength, and  $\phi$  corresponds to the absorbed photon-to-current conversion efficiency at each wavelength ( $\lambda$ ) under illumination of solar light.

The slopes in Fig. 6 depended irradiation intensity I was proportional to the  $\phi$  value for DSCs with electrolytes A–E, which indicated the efficiency of electron injection  $(\Phi_{inj})$  [30]. The higher slopes for DSCs with electrolytes C-E were indicative of the larger efficiency of electron injection ( $\Phi_{inj}$ ) than that for the electrolyte without Lil, but the increase of Lil concentration from 0.1 M to 0.7 M had little influence on the rate of electron injection. Therefore, in terms of the cell performances, the concentration of LiI should be not more than 0.1 M in the electrolyte. When MBI was added to electrolyte C, the slopes decreased from  $143 \text{ mV}^{-1}$  to  $119 \text{ mV}^{-1}$ . The negative shift in the flatband potential of the TiO<sub>2</sub> electrode with MBI would also cause a negative shift in the conduction band edge of TiO<sub>2</sub>, which would decrease the energetic overlap between the sensitizer excited state distribution function and the density of semiconductor acceptor states, and explain the decreased slope or the decreased rate of electron injection for DSCs. Our results indicated that the addition of MBI in the electrolyte could decrease the  $\Phi_{\rm ini}$ , resulting in a decreased  $J_{\rm sc}$ .

#### 3.4. Photovoltaic performance of DSCs

Fig. 7 presents the *J*–*V* curves of DSCs for the samples with A–C and F under 100 mW cm<sup>-2</sup> illumination. It can be seen that the solar cells with electrolyte A showed  $J_{sc}$  of 11.7 mA cm<sup>-2</sup>, an open-circuit voltage ( $V_{oc}$ ) of 0.63 V, fill factor (*FF*) of 0.63 and energy conversion efficiency ( $\eta$ ) of 4.64%. When MBI (0.45 M) was introduced to the 0.7 M DMPII electrolyte (electrolyte A), the  $V_{oc}$  and *FF* of DSCs was obviously improved but  $J_{sc}$  became lower. But when LiI (0.1 M) was introduced to the 0.7 M DMPII electrolyte (electrolyte A), the  $J_{sc}$  of DSCs was obviously improved but  $J_{sc}$  became lower. But when LiI (0.1 M) was introduced to the 0.7 M DMPII electrolyte (electrolyte A), the  $J_{sc}$  of DSCs was obviously improved but  $V_{oc}$  and *FF* became lower. In our experiments, we found that a significant improved  $V_{oc}$  and a slightly enhanced  $J_{sc}$  of DSCs with MBI in the electrolyte (electrolyte B) were obtained in comparison with DSCs using the 0.7 M DMPII electrolyte (electrolyte A) (Fig. 7 and Table 2). This indicated that the addition of MBI in the 0.1 M LiI electrolyte was



**Fig. 7.** *J*-*V* characteristics of the Ru(dcbpy)<sub>2</sub>(SCN)<sub>2</sub>-sensitized solar cells with the electrolytes A–C and F in 100 mW cm<sup>-2</sup> simulated sunlight. Photovoltaic performances of DSCs are indicated in Table 2.

#### Table 2

Photovoltaic performance parameters of dye-sensitized solar cells using electrolytes A-C and F.

Electrolyte	$J_{\rm sc}$ (mA cm <sup>-2</sup> )	$V_{\rm oc} (V)$	FF	η (%)
А	11.7	0.63	0.63	4.64
В	12.2	0.71	0.59	5.11
С	15.5	0.51	0.47	3.72
F	8.0	0.72	0.67	3.86

Measured at an irradiation of AM 1.5.

responsible for the enhanced  $V_{oc}$  and the improved performance of DSCs.

The  $J_{sc}$  for the cells with MBI lower than that for the cells with LiI (Fig. 6) was the result of the decreased  $\Phi_{inj}$  from the excited dyes. MBI in the electrolyte led to a negative shift in  $V_{fb}$  of the TiO<sub>2</sub> conduction band and decreased energetic overlap with the sensitizer excited state donor distribution function, which reduced the electron injection rate so a decreased  $J_{sc}$  was obtained. And the addition of MBI in the electrolyte could increase the  $V_{fb}$  of the TiO<sub>2</sub> electrode and decrease the  $k_{et}$  for  $I_3^-$  reduction at the TiO<sub>2</sub>/electrolyte interface, resulting in the improvement of  $V_{oc}$  and *FF* of the DSCs.

#### 4. Conclusions

In conclusion, the negative shift in  $V_{\rm fb}$  of the TiO<sub>2</sub> electrode by the chemisorbed MBI on the electrode surface could decrease the charge recombination reaction with  $I_3^-$  and also reduce the electron injection yield from the excited sensitizer. From impedance spectra of the solar cells, it can be concluded that the addition of MBI in the electrolyte could decrease the interfacial faradic reaction on the TiO<sub>2</sub> electrode and did not affect the trap distribution in energy below the conduction band. The improved  $V_{\rm oc}$  for solar cells with MBI in the electrolyte resulted from the negative shift in the  $V_{\rm fb}$  of the TiO<sub>2</sub> electrode and the reduction of the rate constant ( $k_{\rm et}$ ) for  $I_3^-$  reduction. Analysis of  $J_{\rm Sc}$  vs. the light intensity for DSCs indicated the higher efficiency of electron injection ( $\Phi_{\rm inj}$ ) for DSCs with various concentrations of Lil in the electrolyte.  $J_{\rm Sc}$  is slightly reduced due to the decreased  $\Phi_{\rm inj}$  after adding MBI to the electrolyte.

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