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Electrolyte effects on photoelectron injection and recombination dynamics in dye-sensitized solar cells

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ABSTRACT

In this study, the influence of electrolyte on the charge recombination, electron injection efficiency (Φ_{ini}) and performance of dye-sensitized solar cells (DSCs) was investigated. A negative shift in the flatband potential (V_{fb}) of the TiO₂ electrode indicated that 1-methylbenzimidazole (MBI) chemisorbed on the TiO2 surface and reduced the Li⁺ adsorption in the nanostructured electrode. Electrochemical impedance spectra for the DSCs showed that the addition of MBI in the LiI electrolyte could increase the electron lifetime and reduce the rate constant ($k_{\rm et}$) for I₃ - reduction with electrons in the TiO₂ conduction band. The increase in V_{oc} for solar cells with MBI was mainly attributed to the negative shift in the V_{fb} of the TiO₂ electrode and the decrease of recombination reaction at the dyed TiO₂/electrolyte interface. Analysis of the short-circuit current density $(J_{\rm sc})$ depended on the light intensity for dye-sensitized solar cells indicated that there was no obvious effect on Φ_{inj} when the LiI concentration was up to 0.7 M in the electrolyte. Due to the negative shift in V_{fb} of the TiO₂ electrode by the addition of MBI in the electrolyte, Φ_{ini} could be reduced to obtain a decreased J_{sc} .

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1. Introduction

Dye-sensitized solar cells (DSCs) developed by Grätzel and co-workers have been extensively studied for their high-power conversion efficiency and potential low cost [\[1–8\]. T](#page-4-0)he typical DSC is normally made into sandwich-type photoelectrochemical solar cells between the dyed $TiO₂$ electrode and the platinized counter electrode, filling with the electrolyte commonly containing the I^-/I_3^- redox couple. Under illumination, photoexcited sensitizers inject electrons into the $TiO₂$ conduction band. The quantum yield of the photoelectron injection depends on the surface properties of TiO2 and the nature and concentration of cations in the electrolyte [\[9\]. T](#page-4-0)he injected electrons can transfer to oxidized dye or I_3^- in the electrolyte resulting in the positive shift of the conduction band. Current strategy to improve the performance of DSCs is related to suppressing the charge recombination and increasing the effective electron injection efficiency.

It has been found that cations in the electrolyte could induce a shift in the flatband potential of $TiO₂$ electrode and affect the rate of electron injection [\[10,11\]. G](#page-4-0)rätzel reported that the flatband potential equals the Fermi level of the nanostructured $TiO₂$ electrode, and is sensitive to the contact electrolyte [\[12\]. R](#page-4-0)edmond and Fitzmaurice investigated cation effects on the flatband potential of the nanostructured TiO₂ electrode by spectroscopic techniques [\[13\].](#page-4-0) They found that the increase of cation content in the electrolyte led to a positive shift in the flatband potential through intercalation into the crystal lattice or chemisorption on the surface of the $TiO₂$ electrode. Meyer and co-workers reported that Li⁺ could accelerate the rate of the electron injection from the photoexcited sensitizers into the TiO₂ conduction band, resulting in a larger short-circuit photocurrent and a lower open-circuit voltage [\[10,11\].](#page-4-0)

Nitrogen-containing heterocycle, such as 1-methylbenzimidazole (MBI) and 4-t-butylpyridine (tBP), is widely used due to the effective improvement on the open-circuit voltage of DSCs [\[14–20\].](#page-4-0) Schlichthorl et al. found that the addition of tBP in the electrolyte resulted in an increased electron lifetime and a significant upward shift of the conduction band edges [\[17\]. H](#page-4-0)agfeldt and co-workers found that the addition of tBP in the electrolyte resulted in a decreased response in the red part of the photocurrent spectrum [\[18,19\].](#page-4-0) They ascribed this effect to a negative shift in the $TiO₂$ conduction band edge. With respect to interface property of the nanostructured $TiO₂$ electrode in the presence of MBI, we reported the negative shift of the flatband potential resulting in a larger increase in the open-circuit voltage. This result implies that MBI interaction with $TiO₂$ could suppress Li⁺ ions on the surface of nano- $TiO₂$ electrode [\[20\]. H](#page-4-0)owever, there is no detailed study about the influence of the shift in the flatband potential of the nanostructured TiO2 electrode on the photoelectron injection from the dye to the $TiO₂$ conduction band and the rate constant for the recombination reaction at the dyed $TiO₂/electrolyte interface$

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Table 1

Composition and concentration of various electrolytes used in this study.

Electrolyte	DMPII, MBI, LiI
A	0.7 M DMPII
B	0.6 M DMPII, 0.45 M MBI, 0.1 M LiI
	0.6 M DMPII, 0.1 M LiI
D	0.4 M DMPII, 0.3 M LiI
E	0.7 M Li
F	0.7 M DMPII, 0.45 M MBI

In this paper, we attempt to study the effect of the negative shift in the flatband potential (V_{fb}) of the nanostructured TiO₂ electrode on the rate constant (k_{et}) for the recombination reaction at the dyed TiO2/electrolyte interface. And we also study the electron injection efficiency (Φ_{ini}) and photovoltaic performance of DSCs with MBI and Li+ in the electrolyte.

2. Experimental

2.1. Materials

1-Methylbenzimidazole (MBI) and lithium perchlorate (LiClO4) were obtained from Aldrich. Anhydrous lithium iodide, iodine, and 3-methoxypropionitrile (MePN) were purchased from Fluka. Acetonitrile (MeCN) (chromatographic grade, Haoshen Chemical, Inc., Shanghai, China) was used without further purification. 1,2- Dimethyl-3-propylimidazolium iodide (DMPII) was synthesized as reported previously [\[15\]. S](#page-4-0)ix different electrolytes were employed for DSCs as shown in Table 1.

2.2. DSC assembly

The colloidal $TiO₂$ nanoparticles were prepared by hydrolysis of titanium tetraisopropoxide as described elsewhere [\[20\].](#page-4-0) Nanocrystalline electrodes about 12 μ m in thickness were obtained by screen-printing $TiO₂$ paste on FTO glass (TEC-8, LOF). The $TiO₂$ film was heated in air for 30 min at 450 \degree C. The film was immersed in an ethanol solution of 0.5 mM cis-dithiocyanate-N,N -bis-(4 carboxylate-4-tetrabutylammoniumcarboxylate-2,2 -bipyridine) ruthenium (II) (N719) for 12 h. The platinized counter electrodes were obtained by spraying H_2PtCl_6 solution to FTO glass followed by heating at 410 ◦C for 20 min. DSCs were fabricated by sealing the counter electrode and the dyed $TiO₂$ film with thermal adhesive films (Surlyn, Dupont). And the electrolyte was filled from a hole made on the counter electrode, which was later sealed by a cover glass and thermal adhesive films.

2.3. Methods

The photovoltaic performances of DSCs with the active area of 4.0 cm2 were measured under an illumination of AM 1.5 (100 mW cm−2), which was realized on a solar simulator (Changchun Institute of Optics Fine Mechanics and Physics, Chinese Academy of Science) with a Keithley 2420 source meter.

The flatband potential (V_{fb}) of the nanostructured TiO₂ electrodes was performed by measuring absorbance at 780 nm as a function of the applied potential. For spectroscopic electrochemistry measurement, 4-µm thick TiO $_2$ film formed the working electrode (2 cm^2 surface area) of a three-electrode photoelectrochemical cell employing a platinum wire counter electrode and an Ag/AgCl reference electrode. Potential control was carried out on a CHI 660A potentiostat, the applied potential being scanned at 5 mV/s. A 780 nm monochromatic light source was obtained from UV–Vis spectrophotometer (TU-1901, PGeneral Instrument Inc., China). For each determination of V_{fb} , a new working electrode and freshly prepared electrolyte solution were used.

Fig. 1. Equivalent circuit used for the impedance spectra of DSCs. R_s : serial resistance: R_{Pr} : the resistance for charge transfer across the Pt/redox electrolyte interface, C_{Pt} : the chemical capacitance (expressed as a constant phase element) at the Pt/redox electrolyte interface, R_{ct} : the resistance for charge transfer across the TiO₂/redox electrolyte interface, C_{μ} : the chemical capacitance (expressed as a constant phase element) of the TiO₂ nanoparticles, $Z_{d(sol)}$: the ions diffusion impedance in the electrolyte.

Steady-state voltammograms were recorded on a CHI 660A electrochemical workstation at room temperature in two-electrode mode of DSCs. Impedance measurements were done using an IM6ex electrochemical workstation (Zahner-Elektrick, Germany) in the frequency range of 50 mHz to 1000 kHz at room temperature. The amplitude of the alternative signal was 5 mV. The impedance measurements were carried out in the dark and the obtained spectra were fitted with Z-View software in terms of appropriate equivalent circuit as shown in Fig. 1 [\[21\].](#page-4-0)

3. Results and discussion

3.1. Effect of additives (1-methylbenzimidazole and LiI) on V_{fb} of the $TiO₂$ electrode

The role of MBI in the electrolyte in determining V_{fb} of TiO₂ electrode was obtained by spectroelectrochemical techniques, measuring the absorbance at 780 nm as a function of the given potential shown in Fig. 2. The value of V_{fb} determined for 0.5 M LiClO₄ in MeCN (-0.74V vs. Ag/AgCl) was 1.47V, which was in accordance with the literature [\[13\].](#page-4-0) The positive shift in V_{fb} observed at 0.5 M LiClO₄ was attributed to the specific adsorption of $Li⁺$ on the TiO₂ electrode. As shown in Fig. 2, the addition of MBI in the electrolyte with 0.5 M LiClO₄ produced a value for V_{fb} of -0.98 V, which was 0.24 V more negative than that for LiClO₄ in MeCN. This negative shift could be attributed to MBI chemisorption with TiO₂ to reduce the adsorption of $Li⁺$ on the electrode surface.

Fig. 2. Absorbance measured at 780 nm as a function of applied potential for a nanostructured TiO₂ electrode in MeCN (0.5 M LiClO₄) and with the addition of 0.5 M MBI. The calculated V_{fb} values were obtained by extrapolating the linear portions of these plots, and the applied potential was scanned at 5 mV/s.

Fig. 3. Nyquist plots of DSCs with electrolytes A–C and F measured at −0.51 V in the dark. The lines show the fitted results. Inset represents enlargement of the area marked with circle.

3.2. Effect of additives on interface recombination in DSCs

Fig. 3 shows the Nyquist plots of DSCs with electrolytes A–C and F measured at forward bias of −0.51 V. The small semicircle in the higher frequencies is due to the counter electrode and could be described by the redox charge transfer resistance (R_{Pt}) , and double layer capacitance (C_{Pt} , expressed as constant phase element). The large semicircle at lower frequencies is ascribed to the electron accumulation and recombination processes in the $TiO₂$ layer, and chemical capacitance (expressed as constant phase element) [\[21–23\]. F](#page-4-0)itting the larger semicircle at lower frequency gives the chemical capacitance (C_{μ}) and the charge transport resistance (R_{ct}) in the EIS experiment. The product of these two quantities gives a time constant, τ = $R_{\rm ct} C_{\rm \mu}$, which corresponds to the observed electron lifetime (τ) [23]. When LiI (0.1 M) was added to the 0.7 M DMPII electrolyte (electrolyte A), the chemical capacitance C_{μ} increased from 2.8 mF to 24.3 mF and the charge transport resistance R_{ct} decreased sharply from 44.2 Ω to 3.5 Ω , yielding the observed τ of 85.1 ms for DSCs with 0.1 M LiI electrolyte (electrolyte C) in comparison with that for DSCs without LiI (electrolyte A), i.e., 123.8 ms. But when MBI (0.45 M) was added to the 0.7 M DMPII electrolyte (electrolyte A), the chemical capacitance C_μ decreased to 0.64 mF and the charge transport resistance $R_{\rm ct}$ increased to 586 Ω , yielding the observed τ of 375 ms for DSCs without LiI in the electrolyte (electrolyte F). It is indicated that a decreased τ for DSCs with 0.1 M Lil electrolyte (electrolyte C) revealed that $Li⁺$ could shift the conduction band of the $TiO₂$ semiconductor downward and increase the charge recombination at the dyed $TiO₂/electrolyte interface$. When MBI was introduced to electrolyte C, the chemical capacitance C_μ decreased to 1.3 mF and the electron transport resistance $R_{\rm ct}$ increased sharply to 452.4 Ω , yielding a longer electron lifetime of 588.1 ms for DSCs with MBI in the electrolyte (electrolyte B). It is clearly seen that the addition of MBI in the electrolyte could decrease the recombination reaction with $I_3^-\,$, yielding an increased τ for solar cells.

To study the trap distribution in the $TiO₂$ electrode and the rate constant k_{et} for charge recombination at the dyed TiO₂/electrolyte interface, EIS was measured at different negative applied potentials for DSC with electrolytes A–C. C_{μ} and R_{ct} have been expressed in the following models [\[21,24–27\].](#page-4-0)

$$
C = C_{a} \exp\left[\frac{-\alpha eV}{k_{B}T}\right] + C_{b} \tag{1}
$$

$$
R_{\rm ct} = R_{\rm ct}^{\rm a} \exp\left[\frac{\alpha_{\rm t} eV}{k_{\rm B}T}\right] + R_{\rm ct}^{\rm b} \tag{2}
$$

where k_B is the Boltzmann constant, T is the temperature, e is the elementary charge, V is the applied potential, C_{a} and R_{ct}^{a} are the

Fig. 4. Results from the impedance data for DSC with electrolytes A–C in the dark at the different applied potentials. (a) Capacitance of DSC with electrolytes A–C. (b) Charge transfer resistance of DSC with electrolytes A–C. The lines indicate the fitted results of C_{μ} using Eq. (1) and $R_{\rm ct}$ using Eq. (2) as fitting models.

pre-exponential factor, C_b is the quasiconstant capacitance at low potentials and R_{ct}^{b} is the lower charge transfer limit, α is a coefficient describing an exponential distribution of trap states below the conduction band edge (α <1), α _t is the transfer coefficient of the faradic reaction.

For a (pseudo) first-order recombination reaction with I_3^- , the reaction rate constant k_{et} can be calculated as expressed in Eq. (3) [\[21\].](#page-4-0)

$$
k_{\rm et} = A \cdot \exp\left(\frac{E_a^0 + \alpha_{\rm t} \cdot F \cdot V}{RT}\right)
$$
 (3)

Here A is the pre-exponential factor, E_a^0 is the activation energy at the potential of 0 V and independent of temperature, and F the Faraday constant. Eq. (3) implies that k_{et} depends on the parameter of α_t and increases with the applied potential at room temperature.

Fig. 4 shows the fitted values of C_{μ} using Eq. (1) and $R_{\rm ct}$ using Eq. (2) as fitting models. The fit yielded α values from Eq. (1) were 0.29 for DSC without LiI (electrolyte A), 0.28 for DSCs with 0.1 M LiI electrolyte (electrolyte C), and 0.24 for DSC with MBI in the electrolyte (electrolyte B), which was in accordance with the literature [\[21,26\].](#page-4-0) This demonstrated that the trap distribution in energy below the $TiO₂$ conduction band was not affected in the presence of MBI. The fit yielded α_t values from Eq. (2) were 0.48 for DSCs without LiI (electrolyte A), 0.64 for DSCs with 0.1 M LiI electrolyte (electrolyte C), and 0.40 for DSCs with MBI in the electrolyte (electrolyte B). From Eq. (3), we concluded that MBI could decrease k_{et} to retard the recombination reaction in the $TiO₂$ electrode, which followed the sequence k_{et} (with LiI/MBI) < k_{et} (without LiI) < k_{et} (with LiI).

Fig. 5. Energy diagram for DSC with electrolyte containing I[−]/I₃− redox. The energy levels of nano-TiO₂, dye, redox, the photoexcitation process, and the electron injection in the conduction band of nano-TiO₂ are shown in the diagram. The solid arrows indicate electron flowways, while the dashed arrows are the possible recombination pathways with the oxidized sensitizer and with the redox couple.

The predominant recombination in DSCs is expected to be the photoinjected electrons reaction with I_3^- ions, due to the fast regeneration of dye by I $\bar{}$ in the electrolyte. And V_{oc} for DSCs with I $^{-}/$ I $_{3}^{-}$ redox electrolyte is also determined by the following equation [\[2,28\]:](#page-4-0)

$$
V_{\rm oc} = \frac{kT}{e} \ln \left(\frac{I_{\rm inj}}{n_{\rm cb} k_{\rm et} [I_3^-]} \right) \tag{4}
$$

where k and T are the Boltzmann constant and absolute temperature, respectively, I_{ini} is the injection electron from dye to semiconductor, n_{cb} the electron concentration at the conduction band edge of semiconductor, and k_{et} the rate constant for the back electron transfer from the conduction band of TiO $_2$ to I_3^- ions. According to Eq. (4), V_{oc} increases with the decreasing of k_{et} . So MBI chemisorption with $TiO₂$ to reduce the adsorption of Li⁺ could improve the V_{oc} for solar cells.

The increased V_{oc} for DSCs in the presence of MBI compared with that in the absence of MBI may be explained by V_{fb} of the TiO₂ photoelectrode. Under Fermi level pinning, these two parameters are linked by the expression [\[28,29\]](#page-5-0)

$$
V_{\text{oc}} = |V_{\text{fb}} - V_{\text{red}}| \tag{5}
$$

where V_{red} is the standard reduction potential of a redox couple assuming that V_{red} does not vary with the addition of MBI. From the energy diagram of the nanocomposite depicted in Fig. 5 in our cell system, it can be seen that the increased V_{oc} for DSCs with the addition of MBI was due to the negative shift of V_{fb} with MBI chemisorption on $TiO₂$ photoelectrode to decrease the adsorption of Li^+ on the electrode surface. From Eqs. (4) and (5), we obtain

$$
\left|V_{\text{fb}} - V_{\text{red}}\right| = \frac{kT}{e} \ln\left(\frac{I_{\text{inj}}}{n_{\text{cb}} k_{\text{et}} [I_{\overline{3}}]}\right) \tag{6}
$$

From Eq. (6), it was found that the decrease of the rate constants k_{et} for I_{3}^{-} reduction at the TiO $_{2}$ /electrolyte interface was attributed to the negative shift in V_{fb} of the TiO₂ electrode. It was suggested that the addition of MBI in the electrolyte could rise up the V_{fb} of the TiO₂ electrode and decrease the k_{et} for I₃ $^-$ reduction at the TiO2/electrolyte interface.

Fig. 6. Short-circuit current $J_{\rm sc}$ as a function of light intensity for DSCs with electrolytes A–E. Inset represents the slopes for electrolytes A–E.

3.3. Effect of additives on photoinjected electrons in DSCs

Fig. 6 shows that the $J_{\rm sc}$ of DSCs with electrolytes A-E varied in direct proportion to the illumination intensity (I) (AM 1.5, 46.4–100 mW cm−2). Because recombination at short circuit is negligible, $J_{\rm sc}$ could be related to the irradiation intensity I by the expression [\[30\].](#page-5-0)

$$
J_{\rm sc} = q\phi A I \tag{7}
$$

where q is the electronic charge, A is a constant and depends on the light harvesting efficiency of dye N719 at each wavelength, and ϕ corresponds to the absorbed photon-to-current conversion efficiency at each wavelength (λ) under illumination of solar light.

The slopes in Fig. 6 depended irradiation intensity I was proportional to the ϕ value for DSCs with electrolytes A–E, which indicated the efficiency of electron injection (Φ_{inj}) [\[30\]. T](#page-5-0)he higher slopes for DSCs with electrolytes C–E were indicative of the larger efficiency of electron injection ($\Phi_{\rm inj}$) than that for the electrolyte without LiI, but the increase of LiI concentration from 0.1 M to 0.7 M had little influence on the rate of electron injection. Therefore, in terms of the cell performances, the concentration of LiI should be not more than 0.1 M in the electrolyte. When MBI was added to electrolyte C, the slopes decreased from 143 mV^{-1} to 119 mV^{-1} . The negative shift in the flatband potential of the $TiO₂$ electrode with MBI would also cause a negative shift in the conduction band edge of $TiO₂$, which would decrease the energetic overlap between the sensitizer excited state distribution function and the density of semiconductor acceptor states, and explain the decreased slope or the decreased rate of electron injection for DSCs. Our results indicated that the addition of MBI in the electrolyte could decrease the Φ_{ini} , resulting in a decreased J_{sc}.

3.4. Photovoltaic performance of DSCs

[Fig. 7](#page-4-0) presents the J–V curves of DSCs for the samples with A–C and F under 100 mW cm−² illumination. It can be seen that the solar cells with electrolyte A showed $J_{\rm sc}$ of 11.7 mA cm⁻², an open-circuit voltage (V_{oc}) of 0.63 V, fill factor (FF) of 0.63 and energy conversion efficiency (η) of 4.64%. When MBI (0.45 M) was introduced to the 0.7 M DMPII electrolyte (electrolyte A), the V_{oc} and FF of DSCs was obviously improved but $J_{\rm sc}$ became lower. But when LiI (0.1 M) was introduced to the 0.7 M DMPII electrolyte (electrolyte A), the $J_{\rm sc}$ of DSCs was obviously improved but $V_{\rm oc}$ and FF became lower. In our experiments, we found that a significant improved V_{oc} and a slightly enhanced J_{sc} of DSCs with MBI in the electrolyte (electrolyte B) were obtained in comparison with DSCs using the 0.7 M DMPII electrolyte (electrolyte A) [\(Fig. 7](#page-4-0) and [Table 2\).](#page-4-0) This indicated that the addition of MBI in the 0.1 M LiI electrolyte was

Fig. 7. J–V characteristics of the $Ru(dcbpy)_{2}(SCN)_{2}$ -sensitized solar cells with the electrolytes A–C and F in 100 mW cm−² simulated sunlight. Photovoltaic performances of DSCs are indicated in Table 2.

Table 2

Photovoltaic performance parameters of dye-sensitized solar cells using electrolytes A–C and F.

Electrolyte	$J_{\rm sc}$ (mA cm ⁻²)	$V_{\alpha c}$ (V)	FF	η (%)
A	11.7	0.63	0.63	4.64
B	12.2	0.71	0.59	5.11
	15.5	0.51	0.47	3.72
F	8.0	0.72	0.67	3.86

Measured at an irradiation of AM 1.5.

responsible for the enhanced V_{oc} and the improved performance of DSCs.

The J_{SC} for the cells with MBI lower than that for the cells with LiI ([Fig. 6\)](#page-3-0) was the result of the decreased Φ_{ini} from the excited dyes. MBI in the electrolyte led to a negative shift in V_{fb} of the TiO₂ conduction band and decreased energetic overlap with the sensitizer excited state donor distribution function, which reduced the electron injection rate so a decreased $J_{\rm sc}$ was obtained. And the addition of MBI in the electrolyte could increase the V_{fb} of the TiO₂ electrode and decrease the k_{et} for I_3^- reduction at the TiO $_2$ /electrolyte interface, resulting in the improvement of V_{oc} and FF of the DSCs.

4. Conclusions

In conclusion, the negative shift in V_{fb} of the TiO₂ electrode by the chemisorbed MBI on the electrode surface could decrease the charge recombination reaction with I_3^- and also reduce the electron injection yield from the excited sensitizer. From impedance spectra of the solar cells, it can be concluded that the addition of MBI in the electrolyte could decrease the interfacial faradic reaction on the $TiO₂$ electrode and did not affect the trap distribution in energy below the conduction band. The improved V_{oc} for solar cells with MBI in the electrolyte resulted from the negative shift in the V_{fb} of the TiO₂ electrode and the reduction of the rate constant (k_{et}) for I3 $^-$ reduction. Analysis of J $_{\rm sc}$ vs. the light intensity for DSCs indicated the higher efficiency of electron injection (Φ_inj) for DSCs with various concentrations of LiI in the electrolyte. J_{sc} is slightly reduced due to the decreased Φ_{inj} after adding MBI to the electrolyte.

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